




**WELDABLE ALUMINIUM ALLOY STRUCTURAL COMPONENT****Publication number:** WO0054967**Publication date:** 2000-09-21**Inventor:** HASZLER ALFRED JOHANN PETER (DE);  
MECHSNER KLAUS ALFONS (DE)**Applicant:** CORUS ALUMINIUM WALZPROD GMBH (DE);  
HASZLER ALFRED JOHANN PETER (DE);  
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Improved shape and strength of the weld in a welded structure are obtained by use of a weldable aluminium product comprising a structural component which is a sheet, a plate or an extruded body and is made of an aluminium alloy containing not more than 1.5 wt % Zn. This component has, adhered on at least one side, a cladding layer made of an AA7xxx-series alloy having a corrosion potential lower than that of said alloy of said structural component. The alloy of the structural component is preferably an AA5xxx-series alloy containing Mg in the range 2 to 6 wt %.

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<b>(21) International Application Number:</b> PCT/EP00/02549 <b>(22) International Filing Date:</b> 17 March 2000 (17.03.00)  <b>(30) Priority Data:</b> 99200831.8                      18 March 1999 (18.03.99)                      EP  <b>(71) Applicant (for all designated States except US):</b> CORUS ALUMINIUM WALZPRODUKTE GMBH [DE/DE]; Carl-Spaeter-Strasse 10, D-56070 Koblenz (DE).  <b>(72) Inventors; and</b> <b>(75) Inventors/Applicants (for US only):</b> HASZLER, Alfred, Jo- hann, Peter [AT/DE]; Auf'm Gräverich 31, D-56179 Val- lendar (DE). MECHSNER, Klaus, Alfons [DE/DE]; Heim- strasse 70, D-56566 Neuwied (DE).  <b>(74) Agent:</b> HANSEN, Willem, Joseph, Maria; Corus Technology BV, P.O. Box 10000, NL-1970 CA IJmuiden (NL).		<b>(81) Designated States:</b> AE, AL, AM, AT, AU, AZ, BA, BB, BG, BR, BY, CA, CH, CN, CR, CU, CZ, DE, DK, DM, EE, ES, FI, GB, GD, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KR, KZ, LC, LK, LR, LS, LT, LU, LV, MA, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, TZ, UA, UG, US, UZ, VN, YU, ZA, ZW, ARIPO patent (GH, GM, KE, LS, MW, SD, SL, SZ, TZ, UG, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, TJ, TM), European patent (AT, BE, CH, CY, DE, DK, ES, FI, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI patent (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG).  <b>Published</b> <i>With international search report.</i>
<b>(54) Title:</b> WELDABLE ALUMINIUM ALLOY STRUCTURAL COMPONENT		
<b>(57) Abstract</b>  Improved shape and strength of the weld in a welded structure are obtained by use of a weldable aluminium product comprising a structural component which is a sheet, a plate or an extruded body and is made of an aluminium alloy containing not more than 1.5 wt % Zn. This component has, adhered on at least one side, a cladding layer made of an AA7xxx-series alloy having a corrosion potential lower than that of said alloy of said structural component. The alloy of the structural component is preferably an AA5xxx-series alloy containing Mg in the range 2 to 6 wt %.		

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## WELDABLE ALUMINIUM ALLOY STRUCTURAL COMPONENT

FIELD OF THE INVENTION

5           The invention relates to a weldable aluminium product comprising a structural component in the form of sheet or plate product or an extruded product, for use as a structural member in a welded structure. Further the invention relates to a welded structure comprising at least one such aluminium product. Such welded structures include marine structures e.g. fixed structures, ships, aerospace vehicles, and land vehicles such as trucks, dump trucks, cars (automobiles) and railway vehicles. While in the following description of this invention reference is mainly made to aluminium sheet or plate products, it is to be understood that products based on aluminium extrusions are also included. As is conventional, the term "aluminium product" refers to aluminium alloy products.

20   DESCRIPTION OF THE PRIOR ART

          When welding aluminium plates or sheets by means of fusion welding, such as MIG, TIG or plasma welding, in particular plates or sheets made of the Aluminium Association AA5xxx-series alloys having Mg as an alloying element in the range of 2 to 6 weight %, for example welding head-to-head by using a V-bevel, a weld bead is formed having a relatively sharp angle between the base

material and the weld toe. In a severe and undesirable case even an undercut may be formed. Usually fatigue cracks start at a so-called hot-spot, which is the notch where the weld bead starts. Because of this relatively sharp angle or notch several organisations setting standards, e.g. for ship-building, indicate that the fatigue behaviour of the welded construction detail is essentially independent of the aluminium alloys used, but is dependent on the construction notch effect. This becomes more apparent for the AA5xxx-series alloys having a relatively high Mg content. For this reason an aluminium alloy having improved strength may still be assessed on its fatigue performance as if it were a less strong aluminium alloy, because of the notch effect. And therefore the acceptable "global stresses" in a welded construction detail are reduced to lower levels than the aluminium alloys used would withstand in themselves.

It is known to improve the conformation of the weld bead zone after welding, e.g. by hammer preening or weld re-melting, but such an extra step should be avoided if possible. There is therefore a need for a product providing an improved weld bead in the welding process.

In the context of disclosure of the invention below, some other prior art documents are here mentioned, though they are not concerned with the problem described above.

WO98/28130 describes providing a clad layer of

filler material on an aluminium core sheet, for the purpose of providing a filler for a welding pool.

In light structures, not assembled by welding together of components, it is known to provide a clad layer on a core sheet, the clad layer having a corrosion potential lower than that of the core sheet. GB-A-1004868 describes materials for use in domestic and industrial water heaters having low corrosion performance, particularly illustrated by an AA6061 alloy sheet clad with an AA7072 alloy layer containing up to 0.01% Cu. Weldability is not mentioned. GB-A-1416134 describes alloys of the AA4xxx-series clad with an AA7072 layer to give corrosion resistance. Formation of tubes of this material by welding is mentioned, but weldability of the material is not discussed.

#### SUMMARY OF THE INVENTION

An object of the invention is to provide a weldable aluminium product for use as a structural member in a welded structure, which gives the welded structure an improved fatigue life.

To achieve this object there is provided in accordance with the invention weldable aluminium product comprising a structural component which is a sheet, a plate or an extruded body and is made of an aluminium alloy containing not more than 1.5 wt% Zn and, adhered on at least one side of said structural component, a cladding layer made of an AA7xxx-series alloy having a

corrosion potential lower than that of said alloy of said structural component.

With this product there is achieved the effect of obtaining after welding a much more smooth transition  
5 from the parent plate or sheet to the weld metal than has been known before in welding of aluminium alloys, in particular of AA5xxx-series alloys. In the weld there is a higher barrier for the crack initiation stage with a resulting increase in fatigue life, and thus a fatigue  
10 limit can be achieved which may be considerably higher than that for welded joints known so far. Further this more smooth transition overcomes the need for additional modification techniques of the weld toe profile, such as local grinding or weld toe remelting techniques. Further  
15 it has been observed that the flow behaviour of the molten metal during welding is improved, resulting in a reduced amount of undesirable inclusions in the weld metal, such as oxides.

The use of the clad layer or layers in the  
20 invention can not only improve the "hot spot" of the welded joint, but may also produce compressive stress at the surface of the welded structure, because of the difference in contraction of the structural component and the clad layer. This may reduce residual tensile stress  
25 due to contraction of the weld bead and thus also influence fatigue behaviour positively.

An additional advantage of welding the clad

aluminium product of the invention is that surface compressive residual stresses may be introduced in the regions where fatigue cracks are likely to initiate, which may extend fatigue life significantly. By applying a cladding on one or both sides of the product additional labourious techniques of introducing residual stresses, e.g. peening, can be overcome.

For certain applications due to the cladding there will be advantages such as visual improvement and better forming (shaping) behaviour because the relatively soft surface avoids crack initiation. And further the weldable cladding allows for higher strength tempers to be used for the alloy of the structural component because of the improved corrosion behaviour due to the anodic protection given by the cladding used.

Although the invention may be applied to all kind of aluminium alloys as the structural member having a corrosion potential higher than that of the cladding applied, the effects found are most pronounced in AA5xxx-series alloys having Mg as alloying element in the range of 2 to 6 weight percent. The amount of Zn is preferably 0.4 to 1.5 weight percent.

Preferably the thickness of the structural component or core is in the range of up to 40 mm, more preferably in the range of 0.5 to 20 mm.

Preferably the thickness of the or each cladding layer is in the range of up to 20% of the thickness of



the structural component. The minimum thickness is preferably 1% of that of the structural component, and more preferably the thickness of the or each cladding layer is 1 to 15% of the thickness of the structural component, and most preferably 1 to 10% of the thickness of the structural component, more particularly 2 to 10%.

Adhesion between the cladding layer or layers and the structural component (core sheet or core plate) is very important to obtain structural integrity after welding. Preferably the adhesion is obtained by rolling. Suitable roll-bonding processes are known. Alternatively the adhesion is obtained by casting a composite ingot having simultaneously cast contacting portions of respectively the materials of the structural component and the cladding layer or layers, followed by rolling of the ingot obtained. This provides a mainly oxide free interface. In the case where the structural component is an extruded body, the cladding layer may be applied by any suitable method, for example roll-bonding.

In one embodiment of the aluminium product in accordance with the invention the structural component is made of an AA5083-series alloy, an AA5059-series alloy, an AA5086-series alloy, or an alloy which is a modification of one of these.

A very advantageous AA5xxx-series alloy for the structural component has the following composition, in weight percent:-

Mg 5.0 - 6.0, preferably 5.0 - 5.6

Mn 0.6 - 1.2, preferably 0.7 - 0.9

Zn 0.4 - 1.5, preferably 0.4 - 1.2

5 Zr 0.05 - 0.25

Cr 0.3 max.

Ti 0.2 max.

Fe 0.5 max.

Si 0.5 max.

10 Cu 0.4 max.

Ag 0.4 max.

Sc 0.5 max.

balance aluminium and inevitable impurities.

In this embodiment with the structural component  
15 clad with a AA7xxx-series alloy a smooth angle between  
the parent component and the weld toe of 130° or more can  
be obtained, which allows for application in welded  
constructions designed for fatigue strength, since with  
the smoother weld groove the fatigue properties of the  
20 welded construction details come close to the welded  
fatigue properties of the parent material, as tested in  
SN-curves.

Another very advantageous AA5xxx-series alloy for  
the structural component has the following composition,  
25 in weight percent:-

Mg 3.0 - 4.5, preferably 3.5 - 4.5

Mn 0.4 - 1.2, preferably 0.4 - 0.75

Zn 0.4 - 1.5, preferably 0.4 - 0.9

Zr 0.05 - 0.25

Cr 0.3 max.

Ti 0.2 max.

5 V 0.2 max.

Fe 0.5 max.

Si 0.5 max.

Cu 0.2 max.

Ag 0.4 max.

10 Li 0.5 max.

Sc 0.5 max.

balance aluminium and inevitable impurities.

Preferably in the aluminium product in accordance with the invention the cladding layer is made of an  
15 AA7xxx-series alloy comprising zinc in a range of 0.3 to 5.0 weight %, more preferably in a range of 0.3 to 2.5 weight %, and more preferably made of an AA7072-series alloy. It has been found that by applying an AA7xxx-series alloy as cladding having zinc in the indicated  
20 range, an improved corrosion protection is provided to the heat-affected zone in particular. The heat-affected zone in Mg-containing alloys in particular has usually the worst microstructure with respect to corrosion resistance due to the high precipitation density, which  
25 limits the application of the welded structure obtained in particular in higher temperature applications, typically above 80°C. Because the cladding remains solid and essentially undisturbed during welding in the region

near the heat-affected zone, after welding a good anodic protection is obtained for the critical heat-affected zone due to the presence of the zinc. Additionally the cladding provides an anodic corrosion protection to the structural component which is not affected due to the welding.

AA7072 alloys have the following composition, in weight percent:-

	Si + Fe	0.7 max.
10	Cu	0.1 max.
	Mn	0.1 max.
	Mg	0.1 max.
	Zn	0.8 - 1.3

balance Al and inevitable impurities.

15 In another aspect of the invention there is provided in an aluminium welded structure having at least two aluminium alloy members joined by welding, at least one of the members being an aluminium product of the invention as described above.

20 Typical welded structures to which the invention is applicable include parts of marine transportation vessels such as catamarans of monohull type, fast ferries, high speed light craft, fixed marine structures, aerospace vehicles, land transportation vehicles such as cars, trucks, lorries and railway vehicles and silos and  
25 armour plate.

Suitable welding techniques which may be applied

include TIG, MIG, laser and electric arc welding, and plasma welding.

Various suitable filler metals may be applied during the welding depending largely on the composition of the structural component of the weldable aluminium product, and include AA5183 and AA5087 in the case where the structural component is made from an AA5xxx-series alloy.

#### INTRODUCTION OF THE DRAWINGS

The invention will be further explained with reference to the drawings in which:-

Fig. 1A represents schematically the weld zone after welding of plates made of typical conventional AA5xxx-series alloy.

Fig. 1B is an enlargement of the circled portion of Fig. 1A at the weld bead.

Fig. 2A represents schematically the weld zone after welding of a product in accordance with this invention.

Fig. 2B is an enlargement of the circled portion of Fig. 2A at the weld bead.

Fig. 3 represents schematically the weld zone as in Fig. 2A of a weld of a product in accordance with the invention, with the difference that in this case four weld passes have been applied.

#### DESCRIPTION OF EMBODIMENTS

In Figs. 1A and 1B there is shown the situation

after welding of non-cladded sheets or plates 2 of  
AA5xxx-series alloy showing a relatively sharp angle  $\alpha$   
between the parent plate or sheet 2 and the weld metal 4,  
which even may give rise to an undercut zone 5; typically  
5 the angle  $\alpha$  is smaller than  $130^\circ$ .

In Figs. 2A and 2B, which also represents the  
situation after welding, it is shown that when welding a  
product 1 in accordance with the invention in which the  
structural component 2 is made of an AA5xxx-series alloy  
10 clad on both sides with an AA7xxx-series alloy layer 3,  
the angle  $\alpha$  between the parent product and the weld metal  
4 is much smoother, which will give rise to an improved  
fatigue life.

In Fig. 3, which also represents the situation  
15 after welding, it is schematically shown that also in a  
case where four welding passes are made, as illustrated,  
when welding a product 1 in accordance with the invention  
in which the structural component 2 is made of an AA5xxx-  
series alloy clad on both sides with an AA7xxx-series  
20 alloy layer 3, the angle  $\alpha$  between the parent product and  
the weld metal 4 is much smoother, which will give rise  
to an improved fatigue life.

#### EXAMPLES

25 The following are non-limitative examples of the  
invention.

Example 1

In an example in accordance with the invention sheets of core alloy forming the structural component of 7 mm thickness having the composition of, in weight percent, 2.56 % Mg, 1.14 % Mn, 0.29 % Zn, 1.11 % Si, 0.46 % Fe, balance aluminium and inevitable impurities are clad by roll-bonding on both sides with AA7072 alloy layers each of 6% thickness of the core alloy and having a corrosion potential lower than that of the core alloy. Pairs of these sheets were welded by means of MIG and TIG. The welding configuration was head-to-head, often also referred to as butt-weld samples. After welding the angle  $\alpha$  between the parent plate and the weld metal was 130° or more and further showed no undercuts, which will result in an increase in fatigue life. In this case, the composition of the alloy of the structural component (core sheet) was a non-standard scrap one, in order to test the welding result.

Example 2

Pairs of unclad AA5083 plates, unclad AA5059 plates and AA5059 plates clad on both sides with an AA7072 layer were butt-welded by MIG using AA5183 filler wire. These plates were made on a laboratory scale. The effects of the clad layer on tensile properties, corrosion resistance and weld geometry have been investigated. The plates are 8 mm thick. In the case of the clad plate,

the thickness of each of the AA7072 clad layers was 2% of the total thickness of the product.

In the welding, the welding speed was 0.7 m/min., the gas was Ar, and three weld passes were made.

5 All plates were of temper H321. The AA5059 alloy and the AA7072 alloy clad had the following compositions:

AA5059 [wt%]

	Si	0.067
	Fe	0.097
10	Cu	0.009
	Mn	0.784
	Mg	5.296
	Cr	0.010
	Zr	0.138
15	Zn	0.550
	Ti	0.023

balance Al and inevitable impurities.

AA7072 [wt%], used as clad layer

20	Si	0.17
	Fe	0.28
	Cu	0.005
	Mn	0.009
	Mg	0.001
25	Cr	0.002
	Zn	1.18
	Ti	0.021



balance Al and inevitable impurities.

The clad layers were applied by roll-bonding.

The properties of the unwelded plates were  
5 established using small Euronorm-specimens and tested  
according to the EN 10002 specification. The cross weld  
tensile specimens have been machined according to DNV  
specification "Rules for Ships", January 1996, Section 1,  
with a gauge length of 32 mm (width of weld + 2 times  
10 plate thickness). Notched bar impact tests were  
performed according to ASTM norm E23 using Charpy impact  
test specimens having a V-shaped notch.

To establish the corrosion performance of the clad AA5059  
material in comparison with unclad AA5059 and AA5083, the  
15 ASSET test (ASTM G66), the SWAAT test (ASTM G85) and the  
EXCO test (ASTM G34) were carried out using specimens  
sensitised up to 25 days at 100°C.

#### Results:-

20 A: Tensile properties of unwelded plate.

To assess properties of the unwelded plates small  
Euronorm specimens were machined in L and LT direction  
and tested according to EN 10002. Two specimens were  
used for each alloy. Table 1 compares the tensile  
25 properties of plate material of AA5083, AA5059 and AA5059  
clad with AA7072. The AA5059 material clad with AA7072  
shows lower values in comparison with the non-clad  
material of 6% in LT- and 10% in L-direction in proof

stress and 5% for both directions in tensile strength.

Table 1: Tensile properties of plate material.

Alloy	LT-direction			L-direction		
	PS [MPa]	UTS [MPa]	Elong. [%]	PS [MPa]	UTS [MPa]	Elong. [%]
AA5083	237/240	348/352	18.4/19.6	256/257	351/351	16.0/16.4
AA5059	264/264	385/383	13.0/15.0	294/294	391/394	14.0/14.0
AA5059/ AA7072	248/248	366/365	18.4/18.4	266/265	369/373	16.2/17.0

5 B: Tensile properties across the weld seam.

Table 2 shows the cross weld tensile properties, which were established according to EN 10002 specification. Specimens according to DNV "Rules for Ships", January 1996, Section 1 were used, having a gauge  
 10 length of 32 mm. Three specimens were used for each alloy; Table 2 gives the average results.

Table 2: Tensile properties of welded specimens.

Alloy	PS (MPa)	UTS (MPa)	A32 (%)
AA5083	140	298	15.8
AA5059	152	301	13.7
AA5059/AA7072	163	312	11.8

15 The comparison of the cross weld properties of AA5083, AA5059 and AA5059 clad with AA7072 shows that in these tests the clad material reaches the levels of 160

MPa in yield and 300 MPa in tensile strength.

Apparently, cladding of AA5059 results in an increase in strength of approximately 10 MPa in both the yield and the ultimate tensile strength.

5

C: Notched bar impact test

Notched bar impact tests were performed according to the ASTM norm E23 using Charpy impact test specimens having a V-shaped notch. The results for the unwelded plate can be seen in Table 3 for the fracture codes L-T and T-L.

10

Table 3: Notched bar impact data for the unwelded plate.

Alloy	Fracture code	Absorbed energy [J/cm <sup>2</sup> ]
AA5083	L-T	20
	T-L	14
AA5059	L-T	21
	T-L	15
AA5059/AA7072	L-T	16
	T-L	16

15

In the welded structures Charpy notch specimens were taken with the notch in the middle of the weld seam and with the notch in the heat affected zone (HAZ). The direction of crack propagation is parallel to the weld seam. The results are listed in Table 4.

Table 4: Notched bar impact data for the welded specimens

Alloy	Notch position	Absorbed energy [J/cm <sup>2</sup> ]
AA5083	weld	31
	HAZ	34
AA5059	weld	33
	HAZ	33
AA5059/AA7072	weld	34
	HAZ	38

## D: Corrosion resistance

*ASSET test*

5           Welded specimens of AA5083, AA5059 and AA5059 clad  
with AA7072 were sensitised at 100°C for 7, 16 and 25  
days and tested according to ASTM G66.

          In this test, the AA5083 material shows no  
corrosion attack when no sensitisation treatment is  
10       applied. After sensitising for 7 days at 100°C and Asset  
testing, pitting grade PA/PB (from N, PA-PC) is observed  
in the AA5083 base material. Further sensitisation up to  
25 days at 100°C causes a concentration of pits grade  
PB/PC in the area where the HAZ meets the base material.

15           The AA5059 material also shows no corrosion attack  
after welding without sensitisation. Sensitising at  
100°C for 7 days does not influence the corrosion  
performance of the AA5059 welded joint. No corrosion  
attack was observed, neither in the weld seam, HAZ nor in  
20       the base material. Sensitisation up to 16 and 25 days

leads to exfoliation in the HAZ having an exfoliation degree EB (from N, EA-ED). The base material is not attacked after a sensitising treatment up to 25 days.

5 The AA5059 material clad with AA7072 shows a dense pattern of small pits (PB) homogenously distributed in both the HAZ and the base material. No exfoliation was found after a sensitisation up to 25 days at 100°C. Since the sensitisation treatment changes the precipitation density in the AA5059 material but does not  
10 influence the AA7072 clad material, the corrosion performance of the AA5059 plate material clad with AA7072 is independent of the sensitisation effect. The corrosion performance of the clad material having a pitting degree PB is comparable with the corrosion  
15 performance of AA5083 after 7 days of sensitising.

#### *SWAAT and EXCO tests*

Beside the ASSET test, which was designed to assess the corrosion susceptibility of 5xxx series alloys, SWAAT  
20 and EXCO tests have been performed to determine differences in the corrosion resistance between the AA5059 with and without clad layer. Welded specimens of AA5083, AA5059 and AA5059 clad with AA7072 were sensitised at 100°C for 7 days and tested for SWAAT  
25 performance according to ASTM G85. Since the EXCO test gives a measure for the exfoliation corrosion susceptibility of 2xxx and 7xxx series alloys, only the

AA5059 clad with AA7072 has been tested using the EXCO procedure. It was established, that there is no significant difference in performance of the AA5083, AA5059 and AA5059 clad with AA7072 in the SWAAt test.

5 The EXCO test gave no indication of severe corrosion attack on the AA7072 clad material.

E: Summary of conclusions.

10 The tensile properties of the AA5059 material clad with AA7072 after welding are approximately 10 MPa higher in comparison with unclad AA5059 alloy material. Before welding the tensile properties of the clad material are slightly lower than the tensile properties AA5059 plate material (this reduction is expected, due to the presence of the clad layers).

In non-sensitised condition after the ASSET test, AA5059 material clad with AA7072 shows pitting of grade PB, whereas the AA5059 material shows no corrosion attack. After a sensitisation of 25 days at 100°C AA5059 shows exfoliation in the HAZ, whereas the corrosion performance of the AA5059 plate material clad with AA7072 is not effected by sensitisation.

25 The corrosion performance of AA5059 material clad with AA7072 is comparable with the corrosion performance of AA5083.

While the invention has been described in conjunction with the exemplary embodiments described

above, many equivalent modifications and variations will be apparent to those skilled in the art when given this disclosure. Accordingly, the exemplary embodiments of the invention set forth above are considered to be  
5 illustrative and not limiting. Various changes to the described embodiments may be made without departing from the spirit and scope of the invention.

**CLAIMS: -**

1. Weldable aluminium product comprising a structural component which is a sheet, a plate or an extruded body and is made of an aluminium alloy containing not more than 1.5 wt% Zn and, adhered on at least one side of said structural component, a cladding layer made of an AA7xxx-series alloy having a corrosion potential lower than that of said alloy of said structural component.

2. Weldable aluminium product according to claim 1, wherein said alloy of said structural component contains Zn in the range 0.4 to 1.5 wt%.

3. Weldable aluminium product according to claim 1 or 2, wherein said alloy of said structural component is an AA5xxx-series alloy containing Mg in the range 2 to 6 wt%.

4. Weldable aluminium product according to any one of claims 1 to 3, wherein said alloy of said structural component is selected from AA5083, AA5059 and AA5086 alloys.

5. Weldable aluminium product according to claim 2, wherein said alloy of said structural component contains, in weight %:



	Mg	5.0 - 6.0
	Mn	0.6 - 1.2
	Zn	0.4 - 1.5
5	Zr	0.05 - 0.25
	Cr	0.3 max.
	Ti	0.2 max.
	Fe	0.5 max.
	Si	0.5 max.
10	Cu	0.4 max
	Ag	0.4 max.
	Sc	0.5 max.

balance aluminium and inevitable impurities.

- 15 6. Weldable aluminium product according to any one of claims 1 to 5, wherein said structural component has two said cladding layers, one on each of two opposite sides thereof.
- 20 7. Weldable aluminium product according to any one of claims 1 to 6, wherein the or each said cladding layer has a thickness which is not more than 20% of the thickness of said structural component.
- 25 8. Weldable aluminium product according to any one of claims 1 to 7, wherein said alloy of the or each said cladding layer is an AA7072 alloy.

9. A welded structure comprising at least two aluminium alloy members joined by welding, wherein at least one of said members is a welded aluminium product according to any one of claims 1 to 8.

10. A welded structure according to claim 9, which is a part of a marine structure, a ship, a land vehicle or an aerospace vehicle.

11. Use of a weldable aluminium product according to any one of claims 1 to 8, as a structural member in a welded structure.

12. Use according to claim 11, wherein the welded structure is a part of a marine structure, a ship, a land vehicle or an aerospace vehicle.

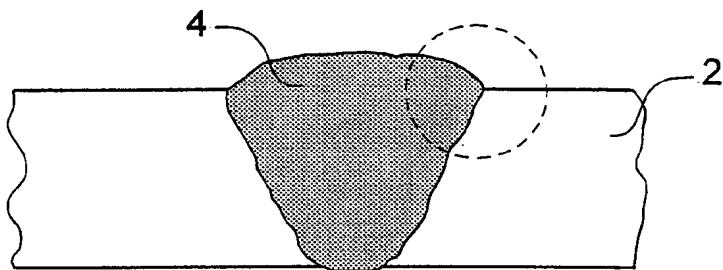


Fig. 1A

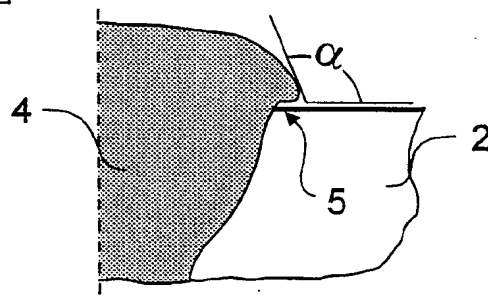


Fig. 1B

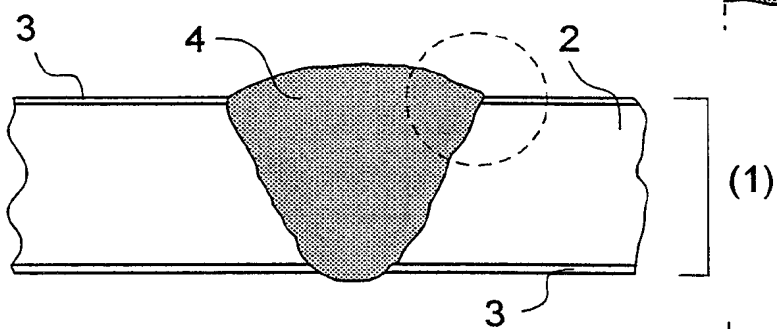


Fig. 2A

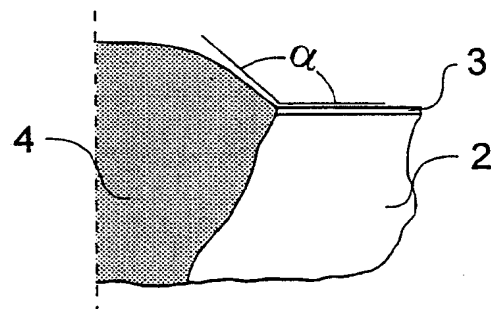


Fig. 2B

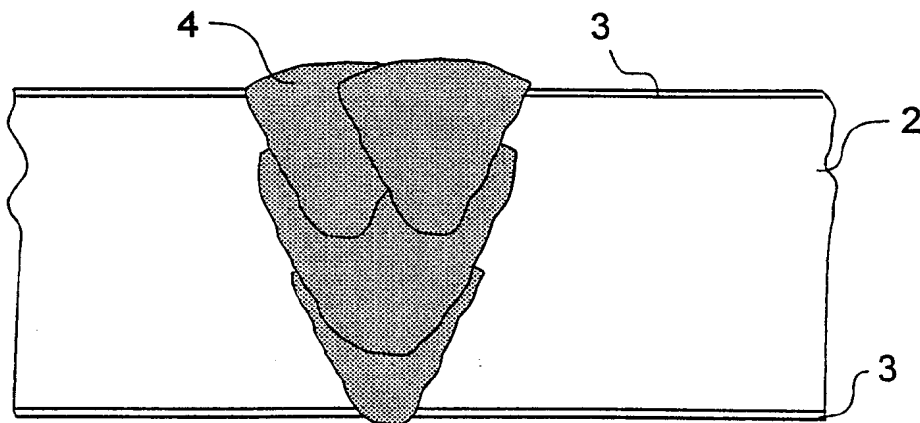


Fig. 3

# INTERNATIONAL SEARCH REPORT

International Application No

PCT/EP 00/02549

**A. CLASSIFICATION OF SUBJECT MATTER**  
IPC 7 B32B15/01

According to International Patent Classification (IPC) or to both national classification and IPC

**B. FIELDS SEARCHED**

Minimum documentation searched (classification system followed by classification symbols)

IPC 7 B32B

Documentation searched other than minimum documentation to the extent that such documents are included in the fields searched

Electronic data base consulted during the international search (name of data base and, where practical, search terms used)

**C. DOCUMENTS CONSIDERED TO BE RELEVANT**

Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	WO 98 28130 A (HOOGOVENS ALUMINIUM WALZPRODUKTE) 2 July 1998 (1998-07-02) cited in the application claims 1-24	1
A	<p>-----</p> <p>DATABASE WPI Section Ch, Week 199514 Derwent Publications Ltd., London, GB; Class M26, AN 1995-101509 XP002139757 &amp; JP 07 024944 A (NIKKEI GIKEN KK), 27 January 1995 (1995-01-27) abstract</p> <p>-----</p> <p style="text-align: center;">-/-</p>	1

☒ Further documents are listed in the continuation of box C.

☒ Patent family members are listed in annex.

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- "A" document defining the general state of the art which is not considered to be of particular relevance
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Date of the actual completion of the international search

8 June 2000

Date of mailing of the international search report

28/06/2000

Name and mailing address of the ISA  
European Patent Office, P.B. 5818 Patentlaan 2  
NL - 2280 HV Rijswijk  
Tel. (+31-70) 340-2040, Tx. 31 651 epo nl,  
Fax: (+31-70) 340-3016

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# INTERNATIONAL SEARCH REPORT

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PCT/EP 00/02549

C.(Continuation) DOCUMENTS CONSIDERED TO BE RELEVANT		
Category *	Citation of document, with indication, where appropriate, of the relevant passages	Relevant to claim No.
A	H.E.MCGANNON: "The Making, Shaping and Treating of Steel" 1971, UNITED STATES STEEL, US XP002139756 page 763, column 1, paragraph 1 ----	1
A	GB 1 004 868 A (REYNOLDS METALS COMPANY) cited in the application *Claims 1-3; page 3, lines 19-49* ----	1
A	GB 1 416 134 A (SWISS ALUMINIUM LTD.) 3 December 1975 (1975-12-03) cited in the application the whole document -----	1

# INTERNATIONAL SEARCH REPORT

information on patent family members

International Application No

PCT/EP 00/02549

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JP 7024944 A	27-01-1995	JP 2956428 B	04-10-1999
GB 1004868 A		NONE	
GB 1416134 A	03-12-1975	NONE	